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Exfoliated graphite with spinel oxide as an effective hybrid electrocatalyst for water splitting†

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The aim of the conducted research was to develop hybrid nanostructures formed from MnCo_2O_4 and exfoliated graphite. Carbon added during the synthesis allowed for obtaining a well-distributed MnCo_2O_4 particle size with exposed active sites contributing to the increased electric conductivity. The influence of the weight ratios of carbon to a catalyst for hydrogen and oxygen evolution reactions was investigated. The new bifunctional catalysts for water splitting were tested in an alkaline medium with excellent electrochemical performance and very good working stability. The results for hybrid samples show better electrochemical performance compared to the pure MnCo_2O_4 . The highest electrocatalytic activity was for sample $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1), where the value of the overpotential was 1.66 V at 10 mA cm^{-2} , and also for this sample a low value of Tafel slope (63 mV dec^{-1}) was denoted.

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In recent years, interest in carbon-based electrocatalysts has increased significantly, which has to do with their unique chemical properties and physical properties which are advantageous for the design of high-performance catalysts.^{1–3} The development of carbon-based catalysts can not only reduce the use of precious metals but also create materials with high surface area, easy functionalization, and chemical stability.^{1,4} In carbon–metal hybrid materials, transition metals exhibit remarkable catalytic abilities, while the carbon material provides better conductivity, greater surface area, prevents aggregation effects, and provides good dispersion of active centers.^{2,5} In addition, interactions between carbon and metallic materials can change the properties of the whole hybrid, contributing to the formation of new active sites with enhanced adsorption capacity.^{3,6} Consequently, many attempts have been made to develop such hybrids as potential candidates for electrochemical applications.⁷ Hybrids based on carbon materials with transition metal sulfides and phosphides have attracted great attention in the hydrogen evolution reaction.⁸ In other studies, an electrocatalyst consisting of iron phosphide

(FeP) and graphene sheets and an electrocatalyst synthesized from nickel phosphide (Ni_2P) nanoparticles and reduced graphene oxide (rGO) doped with nitrogen were developed.^{7,9} Structural characterization showed that high catalytic activity was associated with synergistic effects, promising carbon conductivity, and the presence of electron-rich nitrogen atoms.^{3,7,10}

Transition metal-based spinel-type materials have rapidly become have emerged as promising and highly efficient active materials for potential electrochemical applications due to their numerous advantages such as low prices, abundant resources, and environmental friendliness.^{11–13} Among the wide range of compounds, spinels containing cobalt, nickel or molybdenum are of particular interest as high-performance oxygen evolution reaction (OER) catalysts.^{14,15} One example of the use of bimetallic oxides for HER and OER is NiCo_2O_4 . It is characterized by good activity towards the evolution of oxygen and hydrogen.¹⁶ The presence of transition metal particles such as cobalt and nickel, due to their conductivity and electron configurations, provides a large number of active sites and excellent electrochemical performance in the oxygen evolution reaction.^{12,17} However, other bimetallic oxides, which will also be characterized by good electrocatalytic activity, should also be searched. Recently, electrode materials of spinel-type compounds containing cobalt and manganese have been intensively studied as a very promising electrocatalyst.¹⁸ This material in the form of MnCo_2O_4 is also used for high-performance supercapacitors,¹⁹ lithium-ion batteries,²⁰ photocatalytic CO_2 reduction,²¹ and gas sensors.²² However, to improve the electrochemical performance of spinel structures, a favorable strategy is the introduction of heteroatoms and integration with conducting materials.¹⁸ In order to design a novel and highly active

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MnCo₂O₄/carbon based catalyst, various approaches have been investigated.^{23,24} Only, the synergetic effect of the hybrid nanostructure may have a significant impact on the properties. In one of the strategies, the coupling of MnCo₂O₄ with carbon nanotubes (CNTs) and nitrogen-doped reduced graphene oxide (N-rGO) has been described. The OER performance of the MnCo₂O₄/N-rGO hybrid is comparable to the benchmark iridium-carbon catalyst due to enhanced conductivity and facilitated charge transfer.^{18,25} Another approach proposed the introduction of heteroatoms such as nickel and zinc, followed by their combination with reduced graphene oxide (Mn_{0.4}-Ni_{0.6}Co₂O₄/rGO and Mn_{0.8}Zn_{0.2}Co₂O₄/rGO) is also an effective alternative.²⁶ The hybrid's high performance is mainly attributed to its unique morphology, more exposed active sites, and porous structure with a high specific surface area.^{11,27}

The aim of the conducted research was to examine the influence of the ratios of reagents MnCo₂O₄/exfoliated graphite (EG) in hybrid structures for their catalytic properties. This work aimed to establish effective hybrid composition combining oxides of manganese and cobalt with electrochemically exfoliated cheap, commercially available graphite. The physico-chemical properties of the obtained materials were characterized using different analytical techniques, such as high-resolution transmission electron microscopy (HRTEM), X-ray diffraction (XRD), Raman spectroscopy, and X-ray photoelectron spectroscopy (XPS). The electrochemical properties of hybrid nanostructures were tested in oxygen and hydrogen evolution reactions. It is very important to understand how the ratio of the reagents influences the properties of the final products obtained in the simple synthesis method and the possibility of their use for green hydrogen production.

The morphology of the obtained materials was determined by high-resolution transmission electron microscopy analysis. Fig. 1 on the left side shows the HRTEM images of the synthesized MnCo₂O₄ which have a crystal structure with particle size in the range of 10–30 nm, and the particles tend to agglomerate. Fig. 1 on the right side shows image of the obtained MnCo₂O₄/EG (2/1) hybrid nanostructure which consist of distinct MnCo₂O₄ structures uniformly dispersed on the graphene layer as indicated by the arrows. The hybrid nanostructure exhibit a lower tendency to agglomerate, compared to pure MnCo₂O₄. The HRTEM images for MnCo₂O₄/EG (1/1) and MnCo₂O₄/EG (1/2) in Fig. S1† reveal also that nanostructured MnCo₂O₄ with size with a broad range from 5 to 100 nm among carbon phase them have been successfully synthesised and dispersed in the carbon phase. Additionally, for samples MnCo₂O₄/EG (1/1) and

MnCo₂O₄/EG (1/2) agglomerated particles of MnCo₂O₄ with size more than 200 nm were identified.

X-ray diffraction patterns in Fig. 2(a) confirmed that synthesized MnCo₂O₄ is crystalline with a cubic spinel structure. The Bragg peaks at (111), (220), (311), (400), (422), (511), and (440) were assigned to MnCo₂O₄.²⁸ The intensity of peaks assigned to MnCo₂O₄ decrease with the increasing amount of carbon nanostructure in hybrid materials. An additional prominent peak was determined for MnCo₂O₄/EG hybrid structures located at about the angular value of 26.5°, which could indicate the existence of carbon. For two samples MnCo₂O₄/EG (2/1) and MnCo₂O₄/EG (3/1) with the highest amount of catalyst in the carbon phase additional peaks with low intensity were observed. The presence in these samples of small amount of Co₃O₄ contributed to these additional peaks may have been formed under the conditions of the synthesis procedure.

Raman spectroscopy was used to confirm the presence of carbon in the hybrid nanostructures as well as to check the composition of the prepared samples. Raman spectra for hybrid structures and pure MnCo₂O₄ and exfoliated graphite are present in Fig. 2(b). In all of the hybrid samples with exfoliated

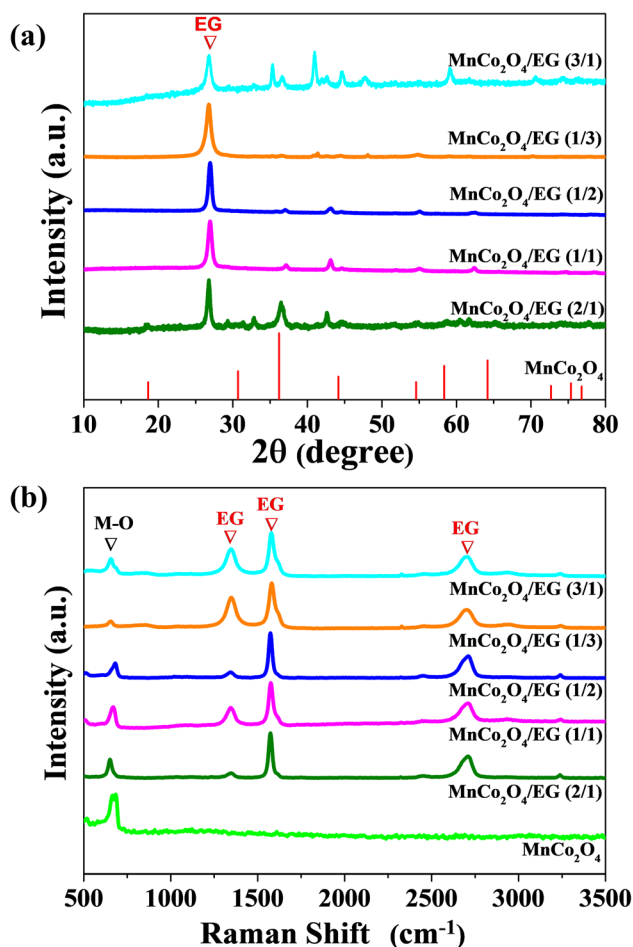


Fig. 2 (a) X-ray diffraction patterns and (b) Raman spectra of MnCo₂O₄ and MnCo₂O₄/EG series.

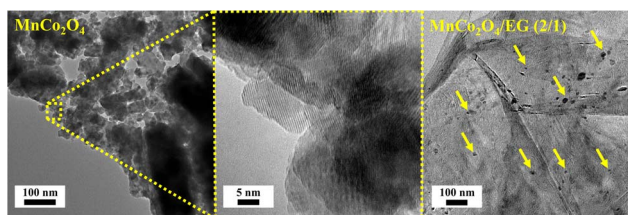


Fig. 1 HRTEM images of MnCo₂O₄ and MnCo₂O₄/EG (2/1).



graphite three prominent peaks at 1347 cm^{-1} , 1580 cm^{-1} , and 2708 cm^{-1} correspond to the D band, G band, and 2D band of carbon, respectively.²⁹ Analysis of intensity ratio of the D band to G band (I_D/I_G) is used to evaluate the disorder of carbon networks.^{30,31} Intensity ratio of I_D/I_G for samples with exfoliated graphite is at the same low level for all samples in the range from 0.30 to 0.41 which indicate that EG do not possess many defected places. For pure MnCo_2O_4 particles, only one characteristic peak at 680 cm^{-1} corresponding to the stretching vibration mode of M–O (M = Mn, Co) was determined.^{32,33} Also, in hybrid samples the peak for M–O is visible. However, the sample is not entirely homogeneous and in some places, a lack of response from M–O vibrations can be observed (see Fig. S2†).

X-ray photoelectron survey spectrum and high-resolution spectra are presented in Fig. 3 for the hybrid sample $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1). Deconvolution of C1s spectra showed the most intense peaks are related to the carbon bonding attributed to C=C (284.4 eV) and C–C (284.9 eV).³⁴ Moreover, other functionalities, such as C–O–C, C–OH (286.3 eV), C=O, O–C–O (287.6 eV), O–C=O (288.5 eV), and π – π (290.1 eV and 292.7 eV)³⁵ formed on the surface of $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) hybrid sample. The formation of functional groups is confirmed by O1s XPS deconvolution obtained for $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1), which reveals the presence of O–Me (529.8 eV), O–Me, O=C (531.6 eV), and O–C–O (532.6 eV). XPS deconvolution of the $\text{Co}2p_{3/2}$ spectrum was fitted with six lines, the first of which lying at a binding energy of 780.1 eV indicates the presence of Co^{3+} ,³⁶ while the line at 781.6 eV is assigned to Co^{2+} .¹⁹ Deconvolution of the $\text{Mn}2p_{3/2}$ spectrum was fitted with five lines, two lines of which lying at a binding energy of 641.4 eV and 643.9 eV indicating the presence of Mn^{2+} (ref. 21) and Mn^{3+} (ref. 37) oxidation sites.

Hybrid materials of spinel oxide with carbon can be an excellent catalyst for water splitting efficiency, due to the potential application of obtained catalysts for hydrogen evolution reaction and oxygen evolution reaction. The linear sweep voltammetry of pure MnCo_2O_4 and obtained hybrid nanostructures was performed using the three-electrode system. The LSV curves at a scan rate of 1 mV s^{-1} in 1 M KOH for OER and

HER were performed. From the LSV curves for OER presented in Fig. 4(a), the OER activity was compared using the required overpotentials to achieve a current density of 10 mA cm^{-2} (η_{10}). For the obtained hybrid samples $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1), $\text{MnCo}_2\text{O}_4/\text{EG}$ (1/1), $\text{MnCo}_2\text{O}_4/\text{EG}$ (1/2), $\text{MnCo}_2\text{O}_4/\text{EG}$ (1/3), and $\text{MnCo}_2\text{O}_4/\text{EG}$ (3/1) the overpotential was 1.66 V , 1.78 V , 1.75 V , 1.79 V , and 1.72 V , respectively. This overpotential was greatly improved for $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) compared with MnCo_2O_4 and even lower than the commercial $\text{RuO}_2/\text{IrO}_2$ catalyst ($\eta_{10} = 1.71\text{ V}$).

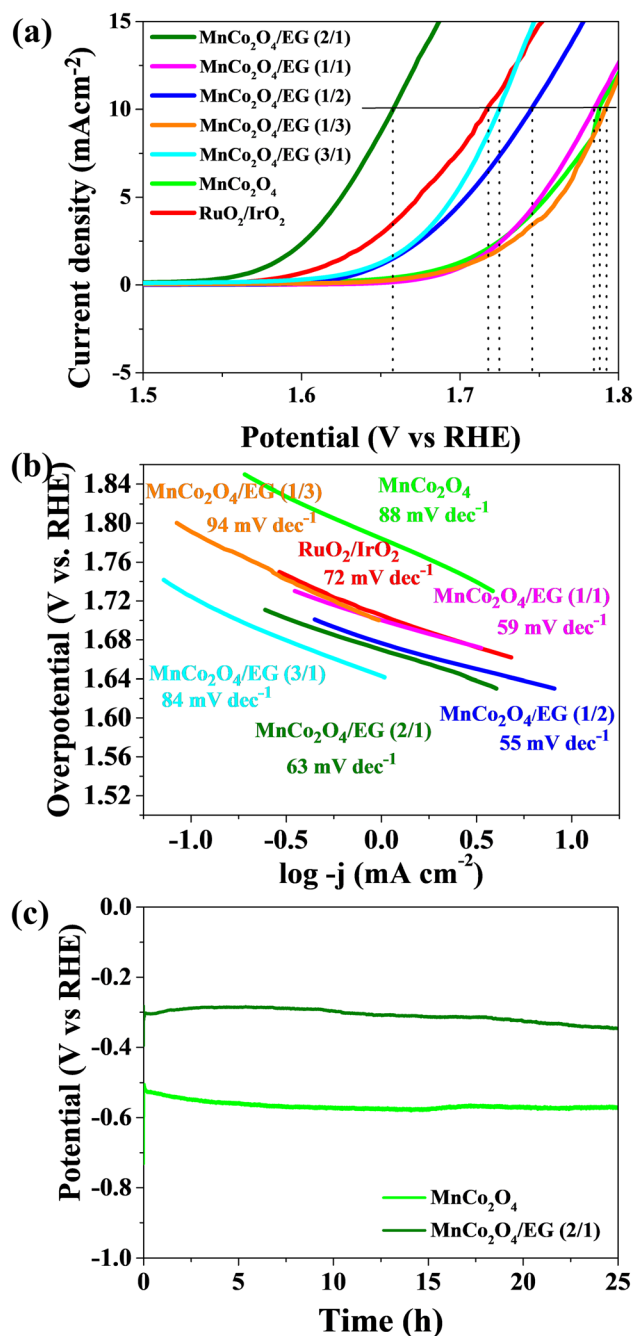


Fig. 4 (a) Linear sweep voltammograms for the OER reaction in 1 M KOH . (b) Tafel plots determined from LSV in 1 M KOH . (c) Stability of $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) and $\text{RuO}_2/\text{IrO}_2$ at a current density of 10 mA cm^{-2} .

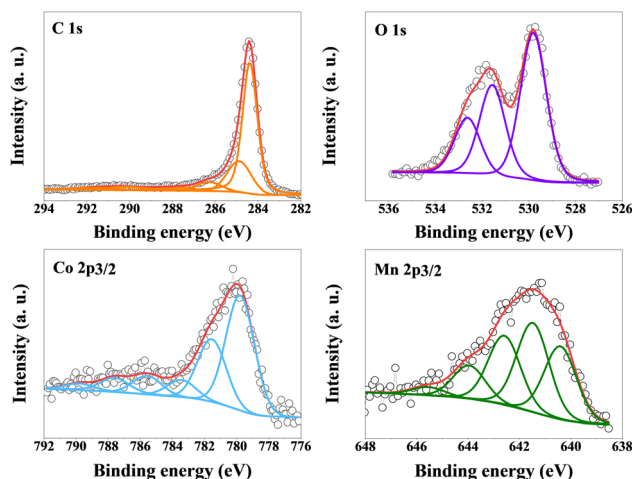


Fig. 3 High-resolution XPS spectra for $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) sample.

Therefore, the highest electrocatalytic activity approximate to commercial $\text{RuO}_2/\text{IrO}_2$ catalyst is shown by sample $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1), for which the value of the potential obtained at the required current density is 1.66 V. This sample has the highest content of MnCo_2O_4 compare other hybrid catalysts. The OER kinetics were estimated using Tafel plots, and the linear region was fitted *via* the Tafel equation: $\eta = a + b \log j$, where η is the overpotential, j is the current density, and b is the Tafel slope. From Fig. 4(b), the calculated Tafel slope for hybrid catalysts is in the range 55–94 mV dec^{-1} , which is superior to that of commercial $\text{RuO}_2/\text{IrO}_2$ (72 mV dec^{-1}) and MnCo_2O_4 (83 mV dec^{-1}). As shown in Fig. 4(c), the addition of exfoliated graphite causes to the high electrochemical stability for sample $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) comparable to commercial $\text{RuO}_2/\text{IrO}_2$.

Besides the OER activity, excellent HER activity is also crucial for bifunctional catalysts. The results of measurements of HER obtained by the linear sweep voltammetry in 1 M KOH for the tested materials are shown in Fig. 5(a). The results for glassy carbon electrode (GCE) have been included in the manuscript for comparison purposes. As can be seen, GCE shows no activity compared to the other catalysts. To evaluate the performance of the obtained catalysts, the potential obtained at a current density of -10 mA cm^{-2} was compared for this electrolyte, and all of the tested samples achieved the required current density. The apparent improvement in the overpotential for the hybrids is evidence that the materials, when combined with exfoliated graphite, show improved performance in HER in 1 M KOH electrolyte than pure MnCo_2O_4 . The polarization curves measured in basic electrolyte show a sharp decrease in the current density as the potential decrease and have an overpotential of -0.39 V , -0.49 V , -0.49 V , 0.58 , and 0.49 for hybrid samples obtained in mass ratio 2/1, 1/1, 1/2, 1/3, and 3/1 (MnCo_2O_4 to EG), respectively.

Among of hybrid materials, sample $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) shows the best catalytic activity and achieved the required current density at a potential value of -0.39 V . Furthermore, this effect is because graphene sheets serve a conducting support, causing enhanced electrical conductivity. This ratio 2/1 of MnCo_2O_4 to EG, where we have lower amount of carbon allows for uniform dispersion of MnCo_2O_4 on the exfoliated graphite surface and obtaining form distribution of particles beneficial in the HER response. The high catalytic activity may have resulted from the high carbonization temperature, which increased the conductivity of the material. From Fig. 5(b), the calculated Tafel slope for hybrid catalysts are in the range of 109–374 mV dec^{-1} , which is still higher than for commercial Pt/C (88 mV dec^{-1}). When we compare these results of hybrids with MnCo_2O_4 (113 mV dec^{-1}) only sample $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) (109 mV dec^{-1}) is superior.

The electrochemical stability of materials is an important parameter which determines commercial application, stability test results are presented in Fig. 5(c). The stability of the hybrid catalyst $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) did not change significantly compared to commercial Pt/C after 24 h, meaning that this material is stable during electrochemical measurements.

The ECSA (electrochemically active surface area) was determined to elucidate the effect of composition on HER and OER. The ECSA was estimated using the CV method's double-layer

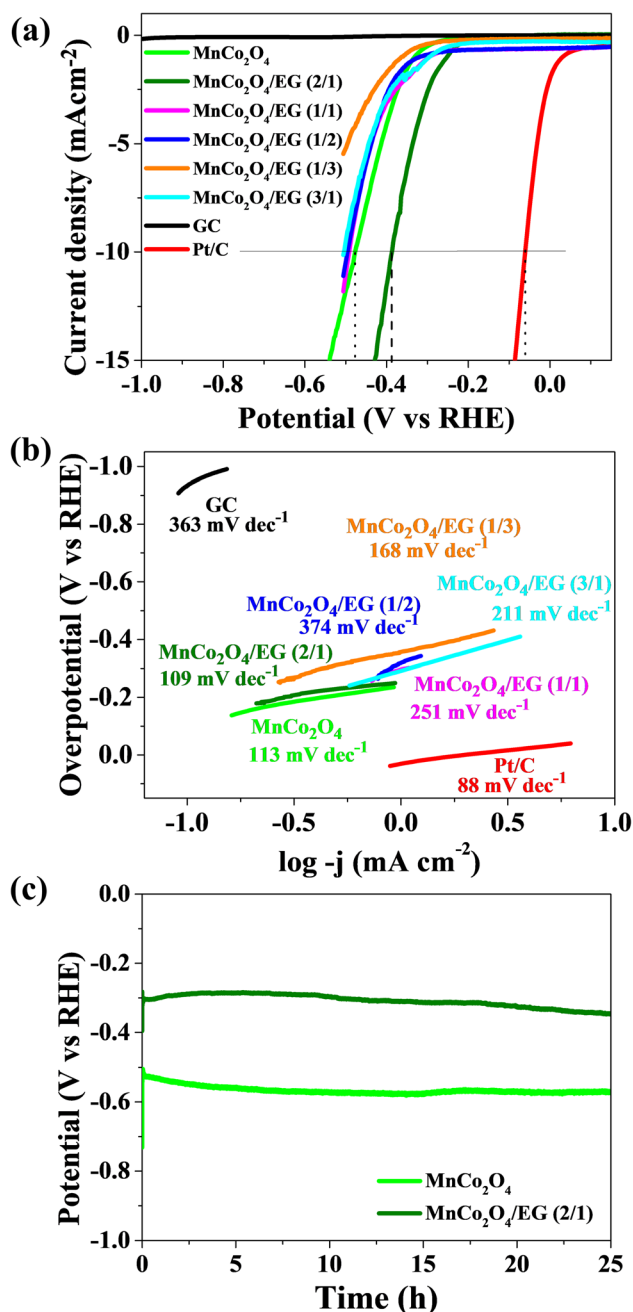


Fig. 5 (a) Linear sweep voltammograms for the HER reaction in 1 M KOH. (b) Tafel plots determined from LSV in 1 M KOH. (c) Stability of $\text{MnCo}_2\text{O}_4/\text{EG}$ (2/1) and Pt/C at a current density of -10 mA cm^{-2} .

capacitance in KOH electrolyte. Fig. S3† shows the CV for obtained catalysts performed in the non-faradaic region at different scanning rates. For the all obtained electrodes in two electrolytes, the charging current at the different scan rates was plotted (Fig. S4†) to obtain the double-layer capacitance from the slope. As can be seen, ECSA differs from the electrolyte used. The ECSA was estimated using the commonly used specific capacity of 0.04 mF cm^{-2} .³⁸ Results are summarized in Table S1.† Although the $\text{MnCo}_2\text{O}_4/\text{EG}$ (1/3) catalyst has by far the highest ECSA, this does not affect the better catalytic properties.



In conclusion, in this case, the ECSA does not have much influence on the catalytic activity.

This work presents a facile production method of MnCo_2O_4 /exfoliated graphite hybrid structures with desired electrochemical properties was proposed. The proposed preparation method leads to the efficient dispersion of MnCo_2O_4 crystallites between carbon sheets with a low tendency to agglomerate of the synthesized nanoparticles, which was confirmed by HRTEM analysis. The obtained spinel MnCo_2O_4 structures on exfoliated graphite were tested in HER and OER reactions. The results indicated that the addition of exfoliated graphite increases the catalytic activity of the MnCo_2O_4 , as well as increases the sample's electron conductivity. The analyzed electrochemical properties in KOH electrolyte for OER have shown that MnCo_2O_4 /EG obtained with a ratio of 2:1 enhanced better catalytic activity because overpotential was lower than for samples obtained with 1:1 and 1:2 ratio of MnCo_2O_4 to EG. The Tafel slope calculated from LSV measured in OER for hybrid catalysts was lower than for commercial $\text{RuO}_2/\text{IrO}_2$ and pure MnCo_2O_4 . The analyzed electrochemical properties in the KOH electrolyte for HER have shown that the addition of carbon for MnCo_2O_4 causes to the increase of overpotential and besides MnCo_2O_4 /EG (2/1) sample remaining analyzed hybrid catalysts possess similar values of overpotential. In summary, the presented new hybrid structures are bifunctional alkaline catalysts, exhibiting excellent electrocatalytic performance and very good working stability for both the HER and OER.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 S. Shit, S. Bolar, N. C. Murmu and T. Kuila, *J. Energy Chem.*, 2021, **59**, 160–190.
- 2 Y. Li, L. Zhou and S. Guo, *EnergyChem*, 2021, **3**, 100053.
- 3 X. Wang, A. Vasileff, Y. Jiao, Y. Zheng and S.-Z. Qiao, *Adv. Mater.*, 2019, **31**, 1803625.
- 4 J. Zhu, L. Hu, P. Zhao, L. Y. S. Lee and K.-Y. Wong, *Chem. Rev.*, 2019, **120**, 851–918.
- 5 X. Peng, C. Pi, X. Zhang, S. Li, K. Huo and P. K. Chu, *Sustainable Energy Fuels*, 2019, **3**, 366–381.
- 6 I. K. Sideri and N. Tagmatarchis, *Chem.–Eur. J.*, 2020, **26**, 15397–15415.
- 7 M. A. Younis, S. Lyu, Q. Zhao, C. Lei, P. Zhang, B. Yang, Z. Li, L. Lei, Y. Hou and X. Feng, *BMC Mater.*, 2019, **1**, 1–26.
- 8 C. Lu, D. Tranca, J. Zhang, F. n. Rodríguez Hernández, Y. Su, X. Zhuang, F. Zhang, G. Seifert and X. Feng, *ACS Nano*, 2017, **11**, 3933–3942.
- 9 S. S. Shinde, A. Sami and J. H. Lee, *ChemCatChem*, 2015, **7**, 3873–3880.
- 10 J. Cao, C. Lei, J. Yang, X. Cheng, Z. Li, B. Yang, X. Zhang, L. Lei, Y. Hou and K. K. Ostrikov, *J. Mater. Chem. A*, 2018, **6**, 18877–18883.
- 11 P. V. Shinde, R. Samal and C. S. Rout, *Trans. Tianjin Univ.*, 2022, **28**, 80–88.
- 12 A. Dymerska, W. Kukulka, M. Biegun and E. Mijowska, *Materials*, 2020, **13**, 3918.
- 13 L. Liu and Y. Xiao, *Comput. Mater. Sci.*, 2021, **187**, 110082.
- 14 Z. Li, M. Hu, P. Wang, J. Liu, J. Yao and C. Li, *Coord. Chem. Rev.*, 2021, **439**, 213953.
- 15 J. Zhou, J. Li, L. Zhang, S. Song, Y. Wang, X. Lin, S. Gu, X. Wu, T.-C. Weng and J. Wang, *J. Phys. Chem. C*, 2018, **122**, 14447–14458.
- 16 X. Gao, H. Zhang, Q. Li, X. Yu, Z. Hong, X. Zhang, C. Liang and Z. Lin, *Angew. Chem., Int. Ed.*, 2016, **55**, 6290–6294.
- 17 Y. Xu, K. Fan, H. Q. Fu, Y. Zou, M. Dong, Y. Dou, Y. Wang, S. Chen, H. Yin and M. Al-Mamun, *Nanoscale*, 2021, **13**, 20324–20353.
- 18 K. Lankauf, K. Cysewska, J. Karczewski, A. Mielewczyk-Gryń, K. Górnicka, G. Cempura, M. Chen, P. Jasiński and S. Molin, *Int. J. Hydrogen Energy*, 2020, **45**, 14867–14879.
- 19 Y. Dong, Y. Wang, Y. Xu, C. Chen, Y. Wang, L. Jiao and H. Yuan, *Electrochim. Acta*, 2017, **225**, 39–46.
- 20 A. K. Mondal, D. Su, S. Chen, A. Ung, H. S. Kim and G. Wang, *Chem.–Eur. J.*, 2015, **21**, 1526–1532.
- 21 S. Wang, Y. Hou and X. Wang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 4327–4335.
- 22 S. Vadivel, G. Balaji and S. Rathinavel, *Opt. Mater.*, 2018, **85**, 267–274.
- 23 C. Sun, J. Yang, Z. Dai, X. Wang, Y. Zhang, L. Li, P. Chen, W. Huang and X. Dong, *Nano Res.*, 2016, **9**, 1300–1309.
- 24 X. Ge, Y. Liu, F. T. Goh, T. A. Hor, Y. Zong, P. Xiao, Z. Zhang, S. H. Lim, B. Li and X. Wang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 12684–12691.
- 25 X.-M. Liu, X. Cui, K. Dastafkan, H.-F. Wang, C. Tang, C. Zhao, A. Chen, C. He, M. Han and Q. Zhang, *J. Energy Chem.*, 2021, **53**, 290–302.
- 26 J. M. Gonçalves, M. N. T. Silva, K. K. Naik, P. R. Martins, D. P. Rocha, E. Nossol, R. A. A. Munoz, L. Angnes and C. S. Rout, *J. Mater. Chem. A*, 2021, **9**, 3095–3124.
- 27 J. Yu, Y. Dai, Q. He, D. Zhao, Z. Shao and M. Ni, *Mater. Rep.: Energy*, 2021, **1**, 100024.
- 28 S. Jayasubramaniyan, S. Balasundari, P. Rayjada, R. A. Kumar, N. Satyanarayana and P. Muralidharan, *J. Mater. Sci.: Mater. Electron.*, 2018, **29**, 21194–21204.
- 29 D. Yang, A. Velamakanni, G. Bozoklu, S. Park, M. Stoller, R. D. Piner, S. Stankovich, I. Jung, D. A. Field and C. A. Ventrice Jr, *Carbon*, 2009, **47**, 145–152.
- 30 A. Guedes, B. Valentim, A. C. Prieto, S. Rodrigues and F. Noronha, *Int. J. Coal Geol.*, 2010, **83**, 415–422.
- 31 F. C. Tai, S. C. Lee, J. Chen, C. Wei and S. H. Chang, *J. Raman Spectrosc.*, 2009, **40**, 1055–1059.



- 32 L.-C. Chen, Z. Chen, Y.-H. Huang, H.-S. Hou, *International Conference on Advanced Materials for Science and Engineering (ICAMSE)*, 2016, pp. 653–656.
- 33 J. Baraliya, *AIP Conf. Proc.*, 2016, 060020.
- 34 C. Moreno-Castilla, M. López-Ramón and F. Carrasco-Marín, *Carbon*, 2000, **38**, 1995–2001.
- 35 J. D. Ding, Y. F. Diao and H. G. Shen, *Adv. Mater. Res.*, 2011, 1211–1214.
- 36 R. Huang, J. Lin, J. Zhou, E. Fan, X. Zhang, R. Chen, F. Wu and L. Li, *Small*, 2021, **17**, 2007597.
- 37 S. G. Krishnan, M. Harilal, N. Arshid, P. Jagadish, M. Khalid and L. P. Li, *J. Energy Storage*, 2021, **44**, 103566.
- 38 P. Connor, J. Schuch, B. Kaiser and W. Jaegermann, *Z. Phys. Chem.e*, 2020, **234**, 979–994.

