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27 Abstract

This work explores a protonated L-proline:glucose (molar ratio 5:1) deep eutectic 28 solvent (DES) in fabricating biopolymer membranes utilizing chitosan (CS). Initially, the 29 miscibility of CS and DES to prepare homogeneous dense blend membranes has been 30 investigated. Different techniques, such as scanning electron microscopy, contact angle 31 (CA), atomic force microscopy (AFM), Fourier transformed infrared spectroscopy (FTIR) 32 and swelling degree (uptake), were used to characterize the structure of the resulting 33 membranes. Within the pervaporation performance for ethanol dehydration, Arrhenius 34 and mass transfer analysis were analysed in detail. Interestingly, the addition of DESs 35 provided superior performance to crosslinked CS: DES membranes compared with the 36 ones lacking DES. Based on the morphology and properties observed, this new concept 37 of CS-based membranes can be alternatively applied in other solvent separations 38 39 requiring hydrophilic membranes.

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Keywords: Proline:glucose, deep eutectic solvents; chitosan; water-ethanol; hydrophilic
pervaporation.

46 **1. Introduction**

A current trend of research deals with the implementation of deep eutectic solvents 47 (DESs), which are an emerging class of sustainable solvents, into new applications and 48 processes. Some examples of applications are as coatings [1], eco-friendly media in 49 reactors [2], extraction of biomolecules from natural products, [3], separation of heavy 50 metal from complex mixtures, CO₂ capture [4], fuel desulfurization [5], assisting 51 chromatographic techniques [6], biodiesel synthesis aided with enzymes [7], chemical 52 and biochemical reactions [8], among others. The importance of DESs relies on their 53 non-toxic character, low cost in synthesis, easy to use, biodegradable, environmentally-54 safe and reusable [9]. As for DES synthesis, these eutectic mixtures tend to be typically 55 formed by linking a hydrogen bond acceptor (HBA), usually quaternary ammonium 56 salts, and a hydrogen bond donor (HBD) [1,10]. As for membrane fabrication, DESs are 57 58 initially utilized as additive agents in DES-supported membranes [11,12], since they can confer a synergistic effect in the separation features of polymer membranes. In fact, 59 such performance improvement showed by DESs in membranes is ascribed to a 60 facilitated diffusion and adsorption of molecules within the functional groups present in 61 DESs [13–15]. Apart from their application as additives, DESs were proposed as pore-62 63 forming additives in phase inversion process via immersion precipitation for synthesizing membranes with asymmetrical patterns [16]. 64

Very recently, our research group has synthetized and implemented a new DESs, based on an amino acid and an organosulfur solute, for fabricating dense chitosan (CS) membranes, which were subsequently applied into the pervaporation (PV) separation of methanol/MTBE mixtures [17]. CS was proposed in this study since it is the most

investigated biopolymer for PV membranes with high performance for the removal of 69 water molecules from less polar or non-polar molecules [18,19], e.g., CS has shown 70 interesting permeation rates and salt rejection in seawater desalination via PV [20]. It is 71 worth mentioning that biopolymer-based membranes (including CS, sodium alginate, 72 cellulose acetate, polylactic acid, among others) are a current scope of study according 73 to the need of replacing chemically synthesized polymers [21,22]. Apart from the 74 75 implementation of DES in biopolymer membranes, the fabrication of mixed matrix membranes (or nanocomposites) through the incorporation of nanomaterials is also a 76 current trend in the field [23–25]. Unfortunately, there is still a big need in finding new 77 DES to assist the fabrication of polymer membranes and improve somehow their 78 physicochemical properties and thus separation performance. It is worth mentioning that 79 the main challenge comprises the rapport of the eutectic mixture and organic phase 80 81 (i.e., polymer) towards their compelling joining. Here, in this research, an original hydrophilic eutectic mixture, like protonated L-proline:glucose at molar ratio 5:1, was 82 used for the unprecedented tailoring of cross-linked CS membranes to beaten the 83 typical issues associated with the suitable distribution of the eutectic mixtures inside the 84 biopolymer. 85

Strategically, the new DES owning hydrophilic character and soluble in water, such as protonated-L-proline:glucose under optimized molar ratio 5:1, has been eventually elaborated [26]. After this, the DES was blended with CS phase to fabricate a homogeneous dope solution, succeeded by an *in-situ* cross-linking via glutaraldehyde (GA). These final membranes were fully characterized and evaluated for practical separation application. PV experiments towards the removal of water from azeotropic water-ethanol model mixtures demonstrated the membranes' appropriateness. Finally, a
model based on the resistances-in-series model allowed us to characterize the mass
transfer phenomenon.

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2. Materials and methods

97 2.1. Chemicals and supplies

As for L-proline, this has been acquired with a purity of at least 98% from Sigma Aldrich.
Glucose (pure, WarChem), hydrochloric acid (analytical reagent, POCH S.A.) and
glutaraldehyde (at 25 wt.% concentration) were obtained and utilized lacking in
additional treatment. The chitosan polymer has been purchased from Sigma Aldrich.

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103 2.2. Eutectic mixture preparation

A protonated-L-proline:glucose (5:1) has been preliminarily prepared. Experimentally,
10 g of L-proline and 86.86 mL of HCl 1M were continuously blended at given
conditions (1000 rpm, 70 °C) up to results in a homogenous and translucent mixture.
Next, glucose (3,13 g) was incorporated into the mixture. Residual water has been later
eliminated via evaporation (BUCHI Rotavapor R-300, V-300 vacuum pump).

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2.3. Membrane fabrication

Overall protocol for membrane preparation is graphically described in **Figure 1**. All chitosan-based membranes have been fabricated using dense-film casting method. To formulate the dope solutions, the right quantity of CS (1.5 wt.%) was dissolved in 2 wt.% acetic acid in water. The polymeric dope mixtures were kept in agitation during a day.

After that, eutectic mixture L-proline:glucose (5 wt.%) has been then incorporated in 115 dope mixtures. Importantly, 5 wt.% of eutectic mixture corresponds to the best ratio for 116 the membranes, agreeing with previous studies [17,27]. The final solution was mixed 117 during 4 h before the application of the *in-situ* cross-linking with GA, which was initially 118 applied to confer a better chemical and solvent stability of the membranes, while 119 ensuring the DES restraint into the polymer membrane [28]. The in-situ cross-linking 120 has been performed using GA (100 μ L), followed by HCl (100 μ L). This final mixture was 121 homogenized during 15 min, and then cast. Membrane samples were subjected to 122 drying at 25 °C over 2 days. In general, the membranes' physical aspect has been 123 found as typical homogeneous and continuous film phase with a thickness ca. 25 µm. 124 Finally, the final membranes were named as follows: cross-linked CS & cross-linked 125 CS:L-proline:glucose, labelled as CS:PRO:GLU. 126



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Figure 1. Preparation strategy of crosslinked chitosan membranes containing L-proline:glucose eutectic mixture.

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131 2.4. Membrane analysis and characterization

Microscopy techniques: SEM and AFM: Structure-morphological property was studied 132 via FEI Quanta FEG 250 SEM microscope. The secondary electron detector was also 133 utilized for the investigation, recording micrographs (at 5 kV accelerating voltage) in 134 high vacuum mode. Preliminarily, the membrane samples were essentially coated via 135 sputtering (gold layer ca. 10 nm), which was applied to balance the scares surface 136 conductivity. The respective micrographs were obtained at proper magnification. 137 Regarding cross-section, membrane specimens were fractured by immersing in liquid 138 N₂. On the other hand, AFM studies were done using Nanosurf EasyScan 2 by contact 139 140 mode using silicon tips (AppNano - SICON series) with 10nN constant force [29].

FTIR analysis: Membrane samples have been analysed via Nicolet iS10 spectrometer
(from Thermo Fisher Scientific) presenting deuterated triglycine sulfate (DTGS)
detector, along with a Golden Gate diamond ATR. Under resolution of 16 cm⁻¹, spectra
data was acquired in the range 4000–400 cm⁻¹.

145 *Contact angle determination:* The surface CA determination has been done with 146 ultrapure water. In this analysis, goniometer OCA15 (Data Physics) was utilized. The 147 data were reported as the average and standard deviation (SD) of at least five assays.

2.4.1. Uptake: As for solvent uptake, which gives the solvent adsorption ability [30], has
been investigated for pure ethanol and different water-ethanol mixtures ranging from 050 wt.% water in ethanol. Membrane sample (approximately 1×5 cm) was first weighed;

after this, it was then subjected to immersion in the binary solvent mixtures (at room
temperature, 2 days). As documented in other investigations [31], the wet membrane
pieces need to be fast cleaned to eliminate the possible residual solvent from surface.
Right away, the weight of the samples was again determined (digital balance, Gibertini,
Crystal 500, Italy). The uptake calculation was done accordingly [32]:

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$$Uptake (\%) = \frac{W_f - W_i}{W_f} \cdot 100$$
 Eq. (1)

where W_f and W_i are the weight values for wet samples after immersion and dry before
 immersion, respectively.

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161 2.5. Pervaporation separation performance

PV tests have been done in a lab operation unit; the schematic illustration and additional information are observed in previous works [33]. In general, an azeotropic water-ethanol (10–90 wt.%, respectively) mixture was poured into a PV cell. The solvent temperature was controlled (20, 30, 40 and 50 °C) and kept constant utilizing a thermo digital circulating bath. At permeate side, the vacuum pressure was controlled at 1 mbar via CRVpro 4 vacuum pump (Welch Vacuum Products, USA).

Membrane samples with an active area of 5.3 cm² were located on a porous metal support at PV cell. Experimentally, permeating vapour was simultaneously condensed and collected by means of glass trap located inside a condenser containing liquid nitrogen. When steady-state was observed, the permeate samples were obtained from for 4 h running test and right away weighted to estimate the total permeate flux (J):

$$J = \frac{Q}{A \cdot t} \qquad \qquad \text{Eq. (2)}$$

where *Q* belong to the weight of the permeate (expressed in kg), *A* is the active membrane area (m²) while *t* is the testing time (h). The partial flux (J_i) has been evaluated as the product of its respective weight fraction (y_i) accordingly [34]:

$$J_i = Y_i \cdot J \qquad \text{Eq. (3)}$$

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179 The separation factor (α) was estimated as follows [34]:

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$$\alpha = \frac{y_{water}/y_{ethanol}}{x_{water}/x_{ethanol}} \qquad \text{Eq. (4)}$$

where *y* and *x* belong to the weight fraction of each compound present in permeate and
feed, respectively. The permeate samples composition was analyzed using Autosystem
XL gas chromatograph with flame ionisation detector (FID) and split/splitless injector
(Perkin Elmer, USA). Separation was done using a 60,0 m x 0,32 mm ID x 1 um (DB624) capillary column (Agilent, USA).

186 The *J* and α data were given as the average of at least three tests to guarantee the 187 results' accuracy.

188 2.6. Mass transfer analysis

To understand the mass transfer in the pervaporation system, the resistances-in-series theory has been used based on the solution-diffusion mechanism [35]. Considering the solution-diffusion model, a mass transfer flux of a component *i* can be denoted as a function of an overall mass transfer coefficient and a driving force, as follows:

$$J_i = K_{overall,i}(P_i^{\circ}\gamma_i x_i - P_P y_i) \quad \text{Eq. (5)}$$

194 Where, $K_{overall,i}$ denotes the total mass transfer coefficient for the component *i*, P_i° is the

vapour pressure for the component *i*, γ_i is the activity coefficient of the component *i*, P_P is the total pressure at the permeate side, x_i is the molar fraction of the component *i* at the liquid side, and y_i is the molar fraction of the component *i* at permeate side. In a pervaporation system, the permeate side is under vacuum pressure so that the total pressure at the permeate side can be neglected, and Eq. (5) can be rewritten and simplified as follows:

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$$J_i = K_{overall,i} \cdot (P_i^{\circ} \cdot \gamma_i \cdot x_i) \quad \text{Eq. (6)}$$

According to García et al. [36] and Arregoitia-Sarabia et. al. [35], overall mass transfer coefficient can be described as the combination for individual resistances

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$$\frac{1}{K_{overall,i}} = \frac{1}{K_{l,i}} + \frac{1}{K_{m,i}} = \frac{P_i^{sat} \gamma_i}{k_{L,i}\rho} + \frac{\delta}{P_i} \quad \text{Eq. (7)}$$

where $k_{L,i}$ is the mass transfer coefficient for the component *i* at the liquid side, ρ is the density of the liquid feed, δ is the membrane thickness, and P_i is the permeability of the membrane for the component *i*.

The mass transfer coefficient has been calculated using the following correlation based on the work of Johnson et. al. [37]:

$$Sh = \frac{k_{L,i} \cdot d_{pervaporation \ cell}}{D_{sol}} = 0,0924 \cdot \left[\frac{\mu}{\rho \cdot D_{AB}}\right]^{0,5} \cdot \left[\frac{R \cdot d^2 \cdot \rho}{\mu}\right]^{0,71} \mathsf{Eq.} \ (8)$$

where $d_{pervaporation cell}$ is the diameter of pervaporation vessel, D_{AB} represents the diffusion coefficient, μ regards the viscosity of the mixture, d is the diameter of propeller, and R is the revolution per second of the propeller.

Finally, the membrane selectivity $(S_{i,j})$ and pervaporation separation index (PSI) are determined as follow:

$$S_{i,j} = \frac{P_i}{P_j}$$
 Eq. (9)

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 $PSI = J(S_{i,j} - 1)$ Eq. (10)

- 218
- 219 **3. Results and discussion**
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3.1. Structure-morphological analysis.

As for prepared membranes either with/without DES, they exhibited a convincing flat 221 222 and smoothie surface with no apparent defects. It was visibly clear that there is no sign 223 of being plastically deformed, this latter is characteristic of dense polymeric membranes [38]. For pristine CS membrane, cross-section view (Figure 2b) showed clear crater-like 224 structure. This is typically observed when deforming via freeze-fracture. Additionally, 225 this typical structure has been also observed in other pristine biopolymer membranes 226 (like chitosan) [28,39]. Referring to CS:PRO:GLU membrane (Figure 2d), it also 227 228 showed a homogeneous dense structure lacking in pores or pinholes between the 229 organic phase and eutectic mixture. Essentially, the obtained morphology is considered as convincing proof of exceptional blending/miscibility among the hydrophilic eutectic 230 mixture and CS, this has been reported in precedented studies, e.g., in 231 polymer/polyethylene glycol blend membranes [40,41]. 232



234 235 **Figur**

Figure 2. SEM surface and cross-section micrographs of the fabricated CS and Lproline:glucose membranes. (a, b) cross-linked chitosan and (c, d) cross-linked chitosan:L-proline:glucose (CS:PRO:GLU).

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Herein, the incorporation of this eutectic mixture has interestingly contributed to obtain a compact but less smooth morphology. As mentioned previously, it has been documented that DESs can act as pore former additive to prepare of porous membranes [16,42]. In this case, our scope is proposed to physically merge and enmesh the chosen hydrophilic solvent in the hybrid membrane for potentialize its separation effect once maintained in the membrane structure. Our hypothesis relies on the tremendously success of DESs for the extraction of low molecular weight molecules
based on their property to tremendously generate H-bonds by means of dipole-dipole
and coulombic forces, among other solute-solvent complexing synergy [43,44].

The eventual success of obtaining homogeneous dense membranes, with convincing 248 evidence of non-DES encapsulation, is due to the synergy of CS at linking some agents 249 thanks to its functional groups. In other words, the lack of functional groups among both 250 251 phases (i.e., polymer and DES) will consequently lead to not enough interacting properties and consequently minimal miscible features [45]. These latter properties 252 become relevant when dealing with the merging of inorganic/organic dispersing agents 253 254 into polymers, where potential presence of functional groups or chemically functionalized nanosized materials eventually achieves good synergy and contact at the 255 256 interface among polymer and filler [46].

257 CS itself presents a large number of NH₂ and OH terminal domains, which makes the biopolymer an exceptional polymer for polymer blend formation [19]. While specific 258 DESs should also possess functional groups to synergistically interact with CS 259 conferring specific properties to the final membrane. Here, it is quite possible that this 260 hydrophilic DES (PRO: GLU), presenting various functional groups (e.g., amino, 261 hydroxyl, oxygen and carbonyl containing groups), could interconnect with CS. Clearly, 262 the surface micrograph of the blend membranes proves a smooth, continuous and 263 defect-free surfaces with lack of any pinholes. In literature, other DES, such as choline 264 chloride-malonic acid, was able to turn the flat surface of chitosan films to a non-265 homogenous morphology while keeping a compact pattern [47]. In this study, we also 266 confirm a smoother surface by adding the DES agent into CS, as observed in **Figure 3**. 267

It is worth mentioning that the DES-free crosslinked CS membrane displayed an average root mean square roughness of $S_q=4.0\pm0.5$ nm, where the incorporation of DES slightly lowered such value to $S_q=3.0\pm0.5$ nm. Certainly, upon the nature of the eutectic mixture, the exaggerated load of the DES may promote a rough structure of the final eutectic-chitosan blend membranes [48].

Crosslinked CS membrane



Crosslinked CS PRO:GLU (5:1)



Figure 3. AFM surface and 3D images (15 \times 15 μ m) of pristine CS and CS:PRO:GLU membranes.

It is worth mentioning that the blending of DES into CS membrane may influence the mechanical and thermal properties of the resultant membranes. In a previous work, it has been stated that eutectic mixtures (e.g., based on L-proline:sulfolane) worsened specific mechanical properties, such as Young's modulus and tensile strength, but such

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an effect was compensate due to the in situ cross-linking [17]. DESs tend to decrease 280 the intermolecular interactions in the CS network, causing the co-called plasticization. 281 The incorporation of plasticizers (like DES) in CS, in fact, produces a transition from a 282 rigid to a softer material with elastic properties [49]. Aside from this effect, DES has 283 been demonstrated to affect the thermal stability of the CS membranes in terms of glass 284 transition temperature (T_g), e.g., DES-modified CS films exhibited a lower T_g in 285 comparison with bare CS membranes. Eventually, this can be ascribed to eutectic 286 mixture decomposition. Interestingly, T_a values decreased proportionally as the DES 287 content increased. According to Jakubowska et al. [47], DES concentration increment 288 results in a decrease in temperature at which decomposition starts. 289

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1 3.2. FTIR and water CA characterization

As seen in Figure 4, the FTIR data essentially confirms the effective merging of the 292 hydrophilic PRO:GLU solvent and the biopolymeric phase. Full spectrum data show a 293 firm and vast patterning with asymmetry ranged approximately 3350-3450 cm⁻¹, which a 294 consequence of overlaying the O-H and N-H stretching oscillations of terminal groups 295 connected by H-bonds. Typically, CS spectrum exhibits a various absorption patterns 296 around 1650 cm⁻¹ (which corresponds to C=O stretching of possible CO=NH₂ terminal 297 group), 1520 cm⁻¹ (which corresponds to N-H wagging of non-acetylated 298 2-aminoglucose) and ca. 1580 cm⁻¹ (which corresponds to N-H wagging of CO=NH₂ 299 300 terminal), which evidenced by other reports [50]. Apart from that, absorption patterning located ~ 1150 cm⁻¹ refer to a skew-symmetric stretching for C-O-C links, 1060 cm⁻¹ and 301 1050 cm⁻¹, ascribed to possibly skeletal frequencies comprising a typical C-O 302 15 303 conjunction. Concerning to the DES, the amino acid (L-proline) owns an extra -NH₂ terminal, identified as an RR'C=NR". Hence, proline is also identified as an imino acid. 304 As proline's three-carbon R-grouping tends to be merged to the alpha-nitrogen terminal, 305 this molecule presents a restrained rigid-ring with rotational behaviour [26]. In the case 306 of glucose, it is a monosaccharide presenting six carbon atoms and an aldehyde group. 307 Considering the typical structure of any DES, both molecules (HBA and HBD) are 308 309 structurally well attached via electrostatic interaction (i.e., H-bonds) to form a final 310 eutectic system [51], this latter interaction is apparently observable with a firm shifting and oscillation for the peak ranged from 3750-1950 cm⁻¹ [52]. When this PRO:GLU 311 312 eutectic system is fused in the CS, this results in modest but clear motion on the usual polymer patterning. This latter interaction reveals an exceptional attraction between 313 phases and is translated to affinity. Clear motions, together with the classic behaviour of 314 the patterning, in the range of 3650- 3100 cm⁻¹, must also be noted showing an 315 overlying of the spectra, which are ascribed to O-H, N-H and C-O oscillations in 316 chemical functionalities of the eutectic mixture and polymer. Jakubowska et al. [47] 317 have also reported similar molecular interactions when prepared and characterized 318 chitosan/DES hybrid materials. Importantly, the resulting reticulated flat films containing 319 glutaraldehyde commonly demonstrate clear absorption increment among 1600-1650 320 cm⁻¹ thanks to N=C bonds [53,54]. The widening at 1550, 1740 and 2850 cm⁻¹ refers to 321 the free aldehyde chain and raised C-H length, respectively. Aliphatic -NH₂ terminals 322 decay directly proportional as the peak 1150 cm⁻¹ decreases. Alternatively, DES have 323 been identified as potential cross-linker agents for CS. Especially, eutectic mixtures 324 containing carboxyl groups are able to interact with the NH₂ groups of CS creating an 325

amide connection [48]. Such an amide connection has been also been observed in CS membranes blended with choline chloride -citric acid [52]. Interestingly, the blending of choline chloride-urea DES into the CS structure was able to create the saccharide ring (C–O–C) between DES and CS, proving the formation of a resilient solid biopolymer membrane without adding cross-linking mediator [55].

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its eutectic mixture-based blend membrane.

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337 A relevant point to highlight during the addition of this hydrophilic DES in membranes

regards its remarkable response on either hydrophilic or hydrophobic nature of the 338 membrane. The DES-free cross-linked chitosan flat membrane revealed angle values 339 of ca. 70° (see Figure 5), which agrees with outcomes published previously, e.g., angle 340 values of chitosan nearly 74-88° [50,56]. In principle, the nature (either hydrophilic or 341 hydrophobic) of the biopolymer indeed stands on its degree of deacetylation, e.g., large 342 deacetylation gives as a result exceptional hydrophilicity in membranes due to many 343 NH₂ terminals tend to be available in the biopolymer [57], this becomes relevant since 344 hydrophilicity is needed for enhanced water adsorption over a membrane interface [58]. 345 Hydrophilic nature of chitosan follows from its polar hydrophilic terminals, including -NH₂ 346 347 and -OH; unfortunately, this polar nature is compromised when applying cross-linking. However, the addition of the PRO: GLU DES in the CS contributed to an enhanced 348 hydrophilicity translated to lower CA values of ca. 50°. This gives an idea that the polar 349 350 groups given by the DES have a meaningful influence at improving the hydrophilic nature of the membrane. Importantly, glucose is classified as polar since hydroxyl 351 groups presents high affinity to hydrogen bonds and effective electronegativity. 352

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Figure 5. CA values pristine crosslinked CS and CS:PRO: GLU membranes.

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3.3. Pervaporation testing

3.3.1. Operating temperature dependence of permeation and separation factor.

PV data for all assayed membranes are reported in **Table S1**. **Figure 6** shows the influence of the feed temperature on the total permeation flux, in which a permeation increase was noted in the range of 20-50 °C in pristine cross-linked and its blend with PRO: GLU DES. It is actually traditional behaviour for polymers as their chains tend to present improved flexibility under high temperature, which consequently contribute to enhanced solvents' sorption. This consequently increases the transport of molecules 372



Figure 6. Permeation and separation factor behaviour in respect to operating
 temperature (10/90 wt.% H₂O/EtOH, pressure: 1 mbar). The semi continuous loops are
 only guiding.

Essentially, an influence of temperature on overall flux was further analysed by meansof the Arrhenius equation, as denoted in Eq. 11.

380

377

$$J = J_o \cdot \exp\left(-\frac{E_{app}}{R \cdot T}\right)$$
 Eq. (11)

In theory, J_o refers to the pre-exponential element, while E_{app} corresponds to the 382 383 apparent activation energy for the transport. The product $R \cdot T$ expresses the typical term of energy. By applying mathematical logarithms in previous Eq. (11), E_{app} is 384 determined directly from the straight line proving a compelling relationship between 385 flux and temperature; it means, an increment in overall flux occurs with temperature 386 increase. As **Table 1** reports, it is noted that water displays lower E_{app} values (ca. 387 5.07 kJ mol⁻¹) in respect to ethanol (~10.27 kJ mol⁻¹) in crosslinked CS membrane, 388 confirming the water affinity of chitosan. On the other hand, the eutectic solvent 389 blending slightly increased the E_{app} in water molecules approximately 5.57 kJ mol⁻¹ for 390 the hybrid CS:PRO:GLU membrane, however, E_{app} value for ethanol has been more 391 impacted by DES incorporation. Particularly, the E_{app} keeps unchanged towards 392 water in respect to ethanol ranging from 20 to 50 °C. It is important to point out that 393 394 the temperature increment impact mainly the water permeation, and it is greatly restricted the permeating ethanol; this is supported by Almeida et al. [60], who 395 documented an improved water solubility in CS films containing DES, in which the 396 water solubility increased when increasing the DES content. In this study, the 397 existence of this hydrophilic PRO: GLU increases the energy demanded for the 398

permeating solvent to be transported over the membrane interface, which is more
prominent for ethanol. This latter aspect is in agreement with the DES hydrophilicity,
which favours for the selective properties for the more polar compounds (like water)
[61].

403

Table 1. Apparent activation energy values for overall flux, and single water and ethanol
 across prepared hybrid materials.

Hybrid formulation	<i>E_{app}</i> (kJ mol⁻¹)		
	Overall	Water	Ethanol
Crosslinked chitosan	5.15	5.07	10.27
Crosslinked CS:PRO:GLU	5.63	5.57	11.76

406

When dealing with selective affinity, α parameter in pure CS membrane decayed when 407 the temperature decreased, as seen in **Figure 6**. Promisingly, separation factor was 408 substantially enhanced via DES incorporation observing data of ca. 1,427 (at 20 ℃). 409 Obviously, large α parameters with lower permeating yield were indeed recorded at the 410 411 minimal testing temperature. It somehow in accordance to the polymer "free-volume" 412 theory, as it declares a thermal movement in chains specifically at amorphous localities boosting an increment in free-volume. It is known that as temperature raises, the 413 414 frequency and magnitude of the chain swing increases provoking a free volume 415 increment [62]. Although the kinetic diameters of water and ethanol are substantially different (2.6 and 4.3 Armstrong, respectively), the thermal movement in polymer chains 416

can indeed facilitate the diffusivity for bigger solvent molecules (like ethanol) over a
membrane interface compromising the separation factor. In addition to this, hydrophilic
eutectic mixtures have demonstrated to break the intermolecular structure of CS and
thus open the network structure, allowing the permeation of solvent molecules [48,60].



Figure 7. Water and ethanol flux in respect to operating temperature (10/90 wt.% $H_2O/EtOH$). The semi continuous loops are only guiding.

Very recently, Jakubowska et al. [47] have reported that the DES addition promotes the 425 free volume increment in polymeric matrices. This is ascribed to the enlargement of 426 free-volume fostering the chain motion, which can produce a decrement for the 427 membrane's selective efficiency. However, an application regarding this hydrophilic 428 PRO: GLU DES did not affect the selective properties of CS membrane. Interestingly, 429 both parameters, such as permeation rate and selectivity, were significantly improved. 430 As shown in Figure 7, DES preferentially promoted the passage of water molecules 431 over the polymeric phase while concurrently hindering a possible ethanol permeation. At 432 this point, the polarity of the solvents becomes relevant for their own transport and 433 434 extraction from complex systems [63].

The analysis for the solvent uptake in the developed membranes is represented in 435 Figure 8. It is clearly observed how both membranes show low solvent uptake when 436 437 there is a minimal concentration of water in the ethanolic solutions. Experimentally, a water concentration increment of the feeding solvent mixture resulted in a higher 438 439 membrane swelling in a water concentration from 10 to 50 wt.%. Surprisingly, the incorporation of PRO: GLU DES provoked a decrement in terms of solvent uptake 440 performance compared with the pristine CS. This latter phenomenon supports 441 Jakubowska's findings [47] in which a DES confers stability in chitosan membranes. 442 443 The uptake is generally expected to be restricted once a crosslinking protocol is implemented as this treatment provides resistance to polymer materials against polar 444 solvents thanks to an enhanced stricture in mobility of chains [64]. Swelling 445 phenomenon is adequately identified as one of the primary bottlenecks of hydrophilic-446 based polymer materials when separating polar molecules [20]. Therefore, less prone to 447

448 be swollen membranes are preferentially needed to acquire a stable separation during449 long-term testing.



450

454

Figure 8. Solvent pptake of CS-eutectic solvent membranes under various water
 percentage in ethanol (at room temperature). The semi continuous loops are only
 guiding.

455 *3.3.2. PV data comparability of cross-linked CS: PRO:GLU membranes with other* 456 *studies*

As for PV testing, it is obvious that the PV separation efficiency of both hybrid and polymer materials is primarily dependent on distinct factors, such as membrane properties (e.g., material type, physiochemical and intrinsic properties, structure, etc.), along with operating parameters including feed composition, operating temperature,

pressure gradient, etc. [34]. Especially, the membrane structure is somehow dictated by 461 the applied membrane preparation strategy [65], while most PV evaluation of 462 membranes has been experimented at different feed concentration and operating 463 parameters. This makes particularly tough to give a fairer comparability among 464 pervaporation results from various works [66]. In our research, we eventually make a 465 comparison of performance outcomes among distinct membrane concepts either 466 unmodified polymer, blends, composites or inorganic tested under close operational 467 parameters, as reported in Table 2. Here, the outstanding selectivity, expressed as 468 separation factor, for crosslinked CS:PRO:GLU membrane was found at 20 °C 469 470 (approximately 1,427), which represents almost 2-folds bigger α value in comparison to unmodified crosslinked chitosan. In addition to this, the highest permeation rates were 471 acquired at the highest tested temperature (at 50 °C) in both membranes while showing 472 473 a decrement in selectivity. The membrane containing the DES showed a slight improvement in permeation (see Table S1). When compared with other reports, 474 crosslinked CS:PRO:GLU membranes exhibited better selectivity than other composite 475 membranes, such as crosslinked PVA-filled GO, CS-filled H-ZSM-5, polyimide (PI)-filled 476 ZIF-8, CS-filled titanium dioxide (TiO₂), PI-filled MSS-1, crosslinked PVA-filled ZIF-8-477 NH₂, among others (see **Table 2**). Depending on the used inorganic fillers filled into the 478 polymer membranes, the aforementioned composites can offer higher permeation rates 479 than our findings. Unfortunately, crosslinked CS:PRO:GLU membranes did not 480 overcome the exceptional selectivity of crosslinked sodium alginate-filled beta zeolite 481 and NaP1 zeolite membranes with unprecedented separation factor values. It is worth 482 pointing out that these membranes (i.e., crosslinked CS:PRO:GLU) are overcoming the 483

484 selective-permeable trade-off of the pristine CS membranes.

486 **Table 2.** Comparison of crosslinked CS: PRO: GLU membrane performance with some

487 composites and inorganic membranes tested with similar water-ethanol mixtures.

Membrane concept	Filler content:	Water %	Testing conditions	J (kg m ⁻² h ⁻¹)	Separation factor	Reference:
			20.90 1			
Crosslinked CS:PRO:GLU	-	10 wt.%	20 ºC, 1 mbar	0.242	1,425	This work
Crosslinked CS:PRO:GLU		10 wt.%	50 ºC, 1 mbar	0.389	831.7	This work
Crosslinked PVA- GO	1 wt.%	10 wt.%	40 ℃, 3 mbar	0.137	263	[67]
CS-filled H-ZSM-5	8 wt.%	10 wt.%	80 ℃, 10 mbar	0.230	152	[68]
Crosslinked sodium alginate- filled beta zeolite	10 wt.%	10 wt.%	30 ℃, 0.6 mbar	0.130	1,600	[69]
Polyimide-filled ZIF-8	12 wt.%	10 wt.%	42 ℃, 44 mbar	0.260	300	[70]
CS-filled TiO ₂	6 wt.%	10 wt.%	80 ℃, 50 mbar	0.340	196	[71]

Polyimide-filled MSS-1	12 wt.%	10 wt.%	42 ℃, 44 mbar	0.310	190	[70]
Crosslinked CS-filled silica	5 wt.%	10 wt.%	70 ℃, 10 mbar	0.410	919	[72]
Crosslinked PVA-filled ZIF-8- NH ₂	7.5 wt.%	15 wt.%	40 ℃, 1 mbar	0.120	200	[73]
NaP1 zeolite membranes	-	10 wt.%	75 ºC, 4 mbar	0.45	200 000	[74]
PVA composite membrane	-	10 wt.%	60 ℃, 5 mbar	0.140	170	[75]
PVA composite membrane (PERVAP 2201, Sulzer Chemtech)	-	10 wt.%	60 ℃, 10 mbar	0.100	100	[76]

490

3.4. Mass transfer performance

491

Through the mass transfer model developed for the pervaporation process, and equations 7 and 8, mass transfer resistances can be determined for the different membranes, which are shown in **Table 3**.

495

Table 3. Mass transfer resistance distribution

496

nembrane $R = l \left(\frac{s}{s}\right) \cdot 10^8$
$\left(\frac{1}{n}\right) 10^8$
0.149 0.149
.219 0.219
0.329 0.329
.502 0.502
113 113
132 132
153 153
209 209

497

498

Table 3 shows how the transport stage in the membrane corresponds to the one that
presents the greatest resistance to mass transfer for the two membranes under study,
as well as for water and ethanol, reaching approximately 99% of the total resistance 30

502 distribution of them. On the other hand, despite the great importance of the resistance value to mass transfer of the membrane, differences in behavior can be observed when 503 comparing the same membrane with water and ethanol, at the same temperature. It is 504 seen that in crosslinked CS membrane, the resistance for ethanol transport is two 505 orders of magnitude bigger compared to water, and as for the membrane modified with 506 DES, a difference of 3 orders of magnitude is obtained, which implies that the 507 508 modification by including DES improves the separation performance, fulfilling the task of 509 obtaining a membrane with a greater hydrophilic character, offering approximately twice the resistance to ethanol than the crosslinked CS membrane, whereas for water, the 510 511 resistances are similar.

512 To compare the performance of membrane modification with DES, it is necessary to 513 compare the selectivity factors, this information is presented in **Table 4**.

514

Table 4. Water/ethanol selectivity for the different membranes.

	Crosslinked CS		Crosslinked CS Crosslinked CS:PRO:GLU	
T (°C)	$S_{W/E}$	$PSI\left(\frac{kg}{m^2h}\right)$	$S_{W/E}$	$PSI\left(\frac{kg}{m^2h}\right)$
20	375	87	756	183
30	290	87	600	181
40	238	80	464	163
50	228	85	416	162

515

517 Regarding the selectivity factors that both membranes present, it is observed how the incorporation of proline-glucose as DES allows a greater selectivity towards water, 518 mainly because they prevent the passage of ethanol molecules through the membrane, 519 which explains the increase in the resistance of the membrane to ethanol doubling its 520 value respect to the resistance towards ethanol in the Crosslinked CS membrane. The 521 addition of PRO:GLU DES to the membrane allows the PSI of the membrane to improve 522 523 more than double in comparison with the crosslinked CS membrane, even the membrane performance is greater than that reported in the literature for the water-524 ethanol system [67]. 525

Finally, the increase in temperature results in selectivity decrement of the Crosslinked
CS:PRO:GLU membrane, being the temperature at 20 °C the one that presents the best
performance, which is corroborated by the PSI value presented by the membrane.

- 529
- 530
- 531

532

534 **4. Conclusions and future research**

This research reveals the fabrication and characterization, for the first time, of dense crosslinked CS-hydrophilic protonated-L-proline: glucose. The membranes present a compelling miscible properties and integration of the original hydrophilic eutectic mixture (PRO:GLU) along the organic biopolymer interface. In general, it was utilized ecofriendly items (including biopolymer, water as primary solvent, "green" eutectic mixture), making the developed hybrid materials as good candidates for fabricating sustainable and eco-friendly dense membranes.

For water-ethanol pervaporation separation, these hybrid membranes offer a 2-fold 542 543 improved pervaporation yield than the DES-free crosslinked chitosan and slight enhancement in permeation. As a perspective, the forthcoming research must be 544 emphasized on enhancing the permeation yield and membranes' selective properties 545 546 when incorporating inorganic phases like nanomaterials. By smartly selecting the hydrophilic nanostructured materials (such as graphene oxide, MXene, UiO-66 MOF), 547 the resultant mixed matrix membranes based on CS may offer unprecedent permeation 548 rates while improving the selective properties as well [23,67]. Also, these membranes 549 can be assayed in other attractive PV separation applications, such as methanol/MTBE 550 [77], water/isopropanol [78], water/hydrazine hydrate [79], requiring hydrophilic 551 552 membranes. Due to their interesting capability to form H-bonding intermolecular forces, DESs could foster an outperforming extraction of some other polar solvents present in 553 azeotropic systems [17,43]. Variety of available DESs, as well as high potential for new 554 developments in this field, tries to design new DES-based membranes having tailored 555 selectivity. Lately, visualizing a possible preparation protocol in a more sustainable way, 556

the crosslinker agent (i.e., glutaraldehyde) could be substituted by another less harmful
substance. Here, green substances, such as cinnamaldehyde [80], genipin [81], could
be an alternative.

560

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570 Conflict of Interest

571 The authors declare no conflict of interest.

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